



Enhanced salt-adsorption capacity of ambient pressure dried carbon aerogel activated by CO₂ for capacitive deionization application

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ABSTRACT

Carbon aerogels (CAs) were prepared by polycondensation of resorcinol and formaldehyde using ambient drying technique and were further activated by CO₂. These were investigated for capacitive deionization (CDI) application. The gel parameters were optimized to obtain mesoporous CA with pore size distribution in 8–10 nm. The CA was further subjected to CO₂ activation at 950°C for different time durations and its effect on surface properties was studied. The surface area of the activated CA increased from 698 to 2057 m²/g with corresponding increase in mesopore area from 289 to 644 m²/g. The pore volume also increased from 0.79 to 1.72 cm³/g. Cyclic voltammetry was performed on the test electrodes and specific capacitance values of 108 F/g was measured at scan rate of 10 mV/s in 1 M NaCl. Electrodes of size 200 × 200 mm were synthesized with an active material loading of ~15 mg/cm² and their CDI performance was evaluated in batch mode experiment. Salt-adsorption capacity as high as 8.4 mg/g for CO₂ activated CA was achieved for the starting electrolyte concentration of 500 mg/L, whereas the adsorption capacity for the CA without activation was 4.1 mg/g. The results suggest that CO₂ activated CA is a good choice for CDI application.

Keywords: Carbon aerogel; CO₂ activation; Capacitive deionization; Ambient drying; Adsorption capacity

1. Introduction

Energy efficient desalination technologies are required to meet the growing scarcity of potable water owing to population increase and industrial development. Capacitive deionization (CDI) is a promising method for removing dissolved salts by electrostatic adsorption reaction. The technique is mainly applicable for brackish water and offers advantage of easy regeneration, low voltage, and ambient operational conditions. CDI process is emerging as an energy-effi-

cient alternative to the established desalination technologies, such as ion exchange, reverse osmosis, electrodialysis, and evaporation. Apart from high-energy efficiency it offers the advantages of long life cycle and non-secondary pollution [1,2].

One of the key factors that has limited scaling up of CDI technology is the availability of electrode material that can be mass produced at low cost and has high salt-adsorption capacity. Significant efforts have been focused on the preparation of various electrode materials to obtain higher CDI performance. Accordingly, different high surface area conducting materials including activated carbon (AC) [3], carbon

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nanotubes (CNT) [4], graphene [5], ordered mesoporous carbon (OMC) [6], and carbon aerogel (CA) [7] and their composites [8,9] have been investigated as electrode material for CDI. Salt-adsorption capacity of electrode material depends on the available surface area and the pore size distribution (PSD) of material. Although for AC the surface area is very large compared to that for CA and CNT, the accessible surface area is low due to dominance of micropores. This offsets the advantage of large surface area and results in comparable salt-adsorption capacity for these materials. Reported values for salt adsorption are 3.68 for AC, 3.32 for CNT, and 3.33 mg/g for CA [10–12]. Even though OMC value of salt adsorption is high 10.56 mg/g [13], these are too expensive from a commercial point of view. Recent studies have been made on composite electrode to utilize the advantage of complementary surface characteristics of high surface area and PSD for improving CDI performance [14,15].

CAs are of interest for CDI application as they have good electrical conductivities and their surface area properties can be tailored by optimizing preparation conditions. CAs are generally synthesized via poly-condensation of formaldehyde with a phenol derivative in aqueous medium followed by supercritical drying [16]. Supercritical drying is costly, time consuming, and requires high-pressure operation. This leads to an uncompetitive price to performance ratio and thus limits commercial viability for large-scale production of CA. Therefore, alternative methods of CA synthesis using ambient drying conditions have been explored to suit the requirement of large-scale production. However, these have limitations of moderate surface area due to suppressed porosity [17–19]. CO₂ activation at elevated temperatures has proven very effective for improving surface area by widening of narrow microporosity and creating new pores in carbon materials [20]. This high surface area can be fully accessible only if there is proper PSD which provides good balance between micro and mesopore. It is well known that catalyzing condition during gel preparation of CA controls the size of interconnected particles and pores in the gel [21]. These need to be optimized for producing CA with high surface area and appropriate PSD that would improve salt-adsorption capacity of CDI electrode.

The objective of the present work was to produce CA with desired porosity by low cost ambient drying method. Resorcinol to catalyst ratio during synthesis was optimized to obtain proper pore size along with high surface area. Textural properties of CA were improved by CO₂ activation. Effect of different CO₂ activation conditions on surface area, pore volume, and PSD was investigated. Electrochemical

performance was studied by cyclic voltammetry and performance of the electrodes prepared with optimized conditions was studied in a CDI test cell.

2. Experimental

2.1. Synthesis of CA

CAs were synthesized by polycondensation of resorcinol (R) and formaldehyde (F) in an aqueous solution in a manner similar to the method used by Lee [22]. RF solution was prepared in water with molar ratio of R/F fixed at 0.5 and weight percent of reactants at 40% in the solution. Sodium carbonate was added as base catalyzing agent (C) to the solution in varying concentration to have different R/C molar ratio. Samples were designated as RFC-100, RFC-200, RFC-300, RFC-400, and RFC-500 for R/C molar ratios of 100, 200, 300, 400, and 500, respectively. The RF solution was gelled for ~70 h in a convection oven at 80°C. After complete gelation water from the gel was exchanged with acetone at regular intervals. The density of the solvent was measured after every exchange to ensure complete water removal. Ambient drying of gel was then done at room temperature for two days. Finally, the dried aerogels were pyrolysed at 850°C in a tubular furnace under inert atmosphere. The pyrolysed sample was treated at 950°C under controlled atmosphere of CO₂ for varying time durations. The CO₂ gas was purged in presence of argon gas with CO₂/Ar ratio of 3:1 at flow rate of ~4 mL/min. The optimum conditions for CO₂ activation temperature and gas flow were achieved by analyzing the balance between improvement in surface properties and material loss. The synthesized sample was collected after it was cooled down to room temperature under argon gas purging. The working electrodes for electrochemical evaluations were prepared using slurry of CA powder with 10 wt% polyvinyl alcohol (PVA) solution in water as binder in weight ratio of 1:3. Glutaric acid as esterifying agent was added to the mixture for cross-linking of PVA to get water-insoluble film [23]. The slurry was spread uniformly on graphite sheet of the size 20 × 20 cm resulting in total active material loading of 6 g for each electrode. The coating was further cured in inert atmosphere at 130°C to obtain CA electrode.

2.2. Characterization for morphology, surface area, and specific capacitance

Scanning electron microscopy (SEM) of CA samples prepared with different R/C molar ratio was carried out using Zigma (Carl Zeiss) system to study their surface morphology. The surface area and PSD

of CA samples with different catalyzing and activation conditions were characterized by analysis of nitrogen adsorption–desorption isotherms measured by ASAP-2020 analyzer (Micromeritics). The samples were degassed at 350°C for 10 h prior to the adsorption measurements. BET method was used for total surface area measurements, and t -plot method was used for estimating mesopore surface area. PSDs were obtained by the BJH method from desorption branch of the isotherms. Total pore volume was calculated from the adsorbed volume of nitrogen at $P/P_0 = 0.99$ (saturation pressure). In order to compare the electro-sorption behavior of CA samples cyclic voltammetry (CV) was performed. This characterization is a preliminary method to investigate the electro-sorption behavior of a material. Measurements were performed with AUTOLAB PGSTAT302 potentiostat in a conventional three electrode cell. The working electrode of size 1.5 cm² was formed using CA material under test as described above. Platinum electrode was used as counter electrode and Ag/AgCl electrode was used as a reference electrode. Measurements were performed in the potential range of –0.5 to +0.5 volts in 1 M NaCl electrolyte in a manner similar to the method used by Wang et al. [5] and Liu et. al. [8]. The samples were subjected to scan rates of 2, 5, 10, and 20 mV/s. The specific capacitance (C) of each electrode was evaluated from CV curves using the relationship $C = I/S\Delta m$, where I (mA) is the average current in the applied potential window, S is applied scan rate in mV/s, and m is mass of active material(mg).

2.3. CDI experiments

Electrosorptive removal of NaCl was measured in batch mode using a continuous recycling system incorporating CDI unit cell, peristaltic pump, a power supply, and a conductivity monitor (Fig. 1). Two electrodes of size 20 × 20 cm were mounted with ~2 mm separation having ~70 mL capacity in CDI unit cell. In a typical experiment 240 mL NaCl solution having an

initial concentration of 500 mg/L was employed as feed solution with flow rate 50 mL/min. A direct voltage of 1.2 V was applied between the two working electrodes. The variation of solution conductivity was monitored by conductivity meter (WTW, Cond 330i). The corresponding concentration could be obtained with a calibration curve made prior to the experiments. Salt-adsorption capacity (Q) defined as salt adsorbed per gram of electrode material was calculated according to the following equation:

$$Q(\text{mg/g}) = (C_i - C_f)V/M \quad (1)$$

where C_i and C_f represent the initial and final concentration in mg/L, V is the volume of the solution in L, and M is the mass of active material in gram [24].

3. Results and discussion

3.1. Morphological examination

Fig. 2(a)–(c) SEM images showing morphologies of the CA obtained at different R/C molar ratios of 100, 300, and 500 (samples RFC-100, RFC-300 and RFC-500). It is seen that CAs are composed of clusters of interconnected carbon nanoparticles of different diameters resulted in continuous porous network. CA derived for RFC-100 (Fig. 2(a)) shows that it has highly dense structure having small particle clusters with mainly micropores. For RFC-300 the image shows network of uniform particles of size ~10 nm with well-developed open and small mesopores. As R/C increases to 500, the particle size and pores size grows further and bigger mesopores dominates.

3.2. Surface area analysis of CAs

Surface area analysis of the synthesized CA by N₂ adsorption showed a strong influence of the R/C molar ratio in textural characteristic (Fig. 3(a) and (b)). It can be seen that the hysteresis in the gas adsorbed vs. relative pressure shifts towards higher pressures with increase in R/C molar ratio indicating the formation of bigger mesopores during gelation process. Sample RFC-100 exhibits a type I isotherm according to Brunnauer–Deming–Deming–Teller classification indicating the predominance of microporosity. The small hysteresis loop in the desorption branch at P/P_0 0.4 shows contribution of mesopores of small size for RFC-200. As R/C molar ratio was increased to 300 the CA displayed a much more developed mesopores structure showing type IV isotherm, with prominent

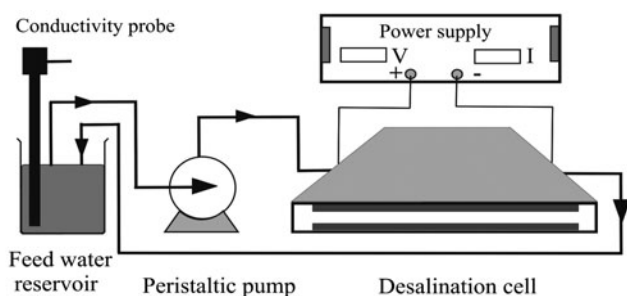


Fig. 1. Schematic diagram of test cell for CDI experiment.

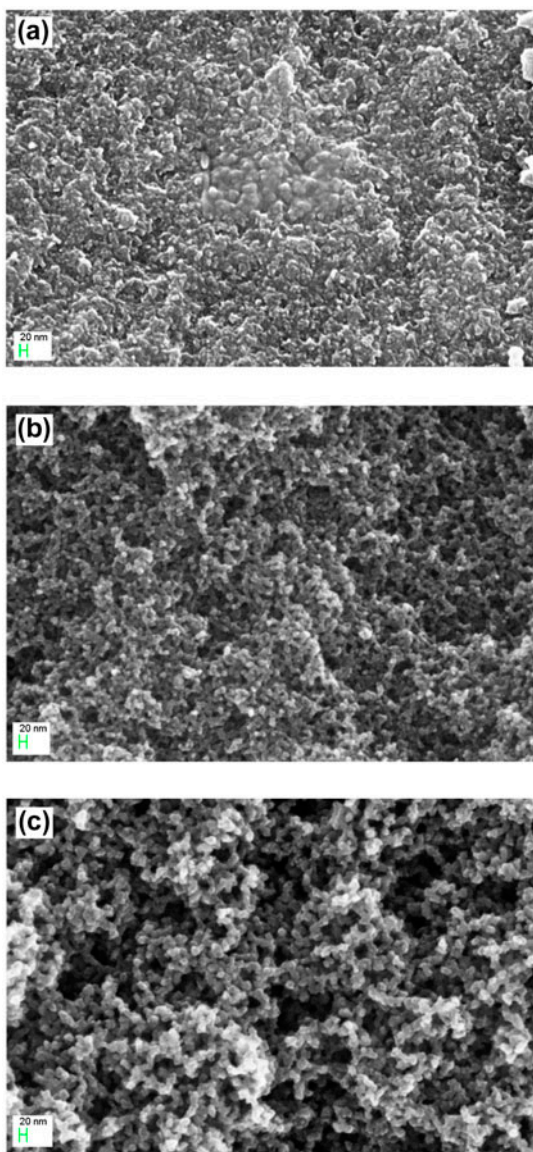


Fig. 2. Scanning electron micrographs of (a) RFC-100 (b) RFC-300 (c) RFC-500.

hysteresis loops at relatively high pressures of P/P_0 0.6. For R/C of 400 and 500 the hysteresis shifts to P/P_0 of 0.7 and 0.8, respectively, indicating presence of bigger mesopores. Fig. 3(b) shows PSD of samples RFC-100, RFC-200, RFC-300, RFC-400, and RFC500 showing that mean pore size increases with the R/C molar ratio. The RFC-100 sample shows the PSD with major contribution of pores below 4 nm diameter. A BJH desorption average pore size lies at 3.6, 8.6, 12.2, and 20.8 for RFC-200, RFC-300, RFC-400, and RFC-500, respectively. These results are in good agreement with the previous works reported in the literature on the porosity development of CA prepared with different R/C molar ratio [21].

Surface area properties of CA prepared with different R/C molar ratio are tabulated in Table 1. Pore volume and average diameter increase with the increase in R/C molar ratio. This behavior is expected because with increasing R/C molar ratio, the catalyst content reduces, resulting in formation of bigger pores. For R/C from 100 to 300, it is observed that both surface areas and pore volume are increasing. But for RFC-400 and RFC-500, although pore volume is increasing because of the presence of bigger mesopores >10 nm, the surface area is decreasing. It emerges that among the prepared samples, RFC-300 has sufficiently high pore size to overcome diffusional limitation of ions as well as highest total surface area and mesopore surface area (698, 289 m²/g).

3.3. Effect of CO₂ activation on the porous structure of CA

Among the samples prepared with different R/C molar ratio, RFC-300 has highest total as well as mesopore surface area and the average pore size is ~8.6 nm which is sufficient for fast ion kinetics inside pores. The surface area properties of this sample were improved by CO₂ activation to achieve better salt-adsorption capacity. CO₂ activation was carried

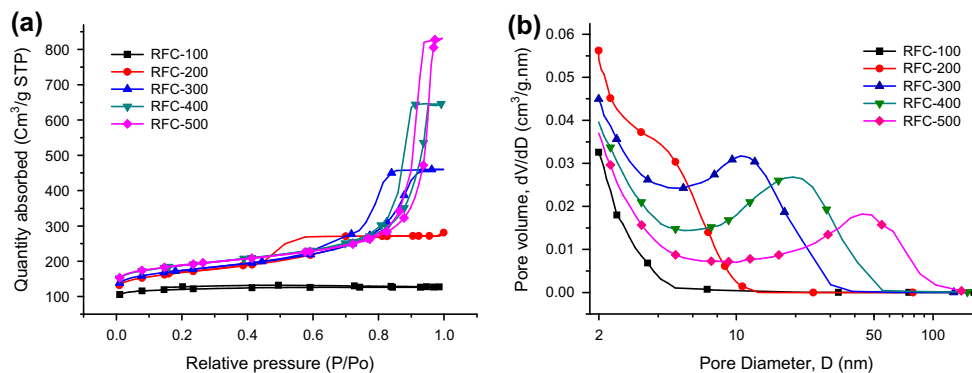
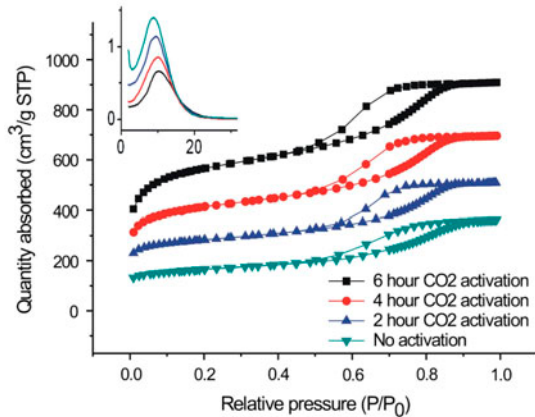


Fig. 3. (a) Adsorption isotherms at 77 K and (b) PSD of CA with different R/C molar ratios.

Table 1

Surface characteristics of CA samples prepared with different R/C molar ratio

Sample	S_{BET} (m^2/g)	S_{Meso} (m^2/g)	Pore V_{total} (cm^3/g)	Pore D_{av} (nm)
RFC-100	490	52	0.21	2.4
RFC-200	604	210	0.39	3.6
RFC-300	698	289	0.79	8.6
RFC-400	628	224	1.13	12.2
RFC-500	601	178	1.33	20.8

Fig. 4. CO_2 activation isotherms with inset showing $dV/d\log(D)$ vs. pore diameter D (nm) for RFC-300 with different activation duration.

out at 950°C with varying activation time and corresponding N_2 isotherms are shown in Fig. 4. The surface properties of CO_2 activated CA are summarized in Table 2. Activation causes improvement in surface area accompanied by material loss due to burn-off. The isotherms show a steep increase in the amount of gas adsorbed at low P/P_0 indicating that large quantity of micropores has been introduced after activation. This might be due to the creation of new micropores by oxidation of certain structural components and opening of previously inaccessible pores. The isotherms are shifted upwards and area under hysteresis is also marginally increased with activation time which indicates that there is an increase in the area of both micro and mesopores. For all the isotherms shown in Fig. 4 hysteresis loops are located almost

over the same P/P_0 range indicating that activation process does not affect the size of pores and PSD significantly. The surface area was found to increase from 698 to $2057 \text{ m}^2/\text{g}$ (mesopore surface area from 289 to $644 \text{ m}^2/\text{g}$) after 6 h activation. It can be observed that although the average diameters of pores with different activation times are slightly decreasing, there is a significant increase in pore volume. For higher activation duration, percentage increase in surface area is overtaken by material burn-off hence 6 h activation was the optimum. The changes observed in the porous structure of the studied samples are reflected in difference in their electrochemical performance.

3.4. Specific capacitance measurements

The CV profiles of the electrodes prepared with CA synthesized at different R/C molar ratio measured at a scan rate of 10 mV/s were displayed in Fig. 5. The closed area of CV curve is an indicator of capacitance of the material. Profiles show quasi-rectangular symmetric and reversible shape in the voltage range -0.5 to 0.5 V . There are no additional oxidation/reduction peaks observed in the chosen potential range, suggesting the ideal capacitive behavior. It is observed that as we proceed from RFC-100 to RFC-500, the CV curve approaches towards rectangular shape which is due to fast ion kinetics for bigger pores. For RFC-500 although the shape is nearly rectangular but the capacitance is lower as bigger mesopores diminishes total surface area. The maximum specific capacitance of 61 F/g was obtained

Table 2

Surface properties of RFC-300 with CO_2 activation

Activation duration	S_{BET} (m^2/g)	S_{Meso} (m^2/g)	Pore V_{total} (cm^3/g)	Pore D_{av} (nm)	% Burn-off
No activation	698	289	0.79	8.6	6
2 h	1,142	424	0.91	7.9	15
4 h	1,647	547	1.01	7.0	25
6 h	2057	644	1.72	6.6	40
8 h	2,298	705	1.89	6.3	70

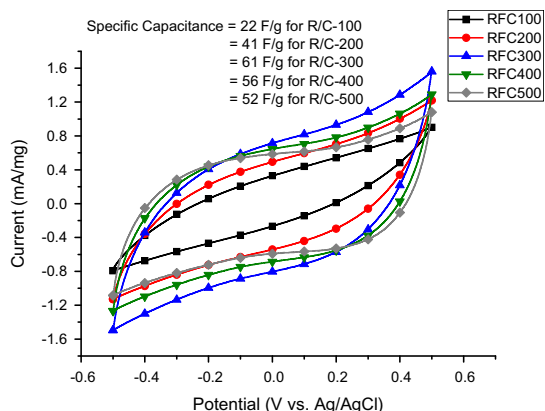


Fig. 5. CV curves for samples with different R/C molar ratio at 10 mV/s scan rate.

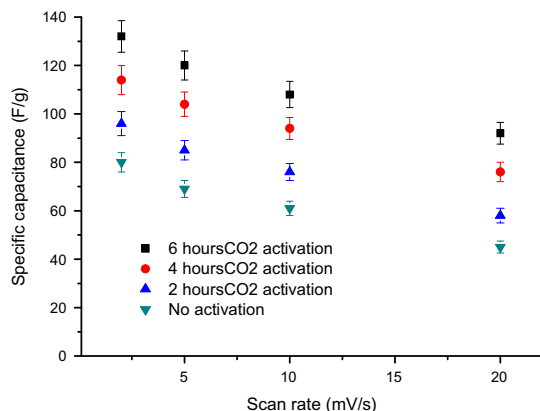


Fig. 6. Specific capacitance of RFC-300 with varying CO₂ activation time at different scan rates.

for RFC-300 which has highest total as well as meso-pore surface area among the selected samples.

RFC-300 was further activated by CO₂ for different hours and its specific capacitance was evaluated. The

calculated specific capacitance values obtained at different scan rates from CV curve are plotted in Fig. 6. It is observed that significant improvement in capacitance was obtained with increasing activation duration as increase in activation leads to improvement in surface area and pore volume of CA. The decrease in capacitance for all the samples with scan rate is not very pronounced as for selected sample PSD is sufficiently high to overcome diffusional limitation. Specific capacitance values for untreated RFC-300 at scan rate of 2, 5, 10, and 20 mV/s were 80, 69, 61, and 45 F/g and after CO₂ activation for 6 h, the values significantly improved to 132, 120, 108, and 92 F/g, respectively. The values are higher than those reported for the ambient dried CA [22]. The higher capacitance values imply higher efficiency in removing salt ions by electrosorption desired for better desalination performance.

3.5. CDI performance

The CDI behaviors of electrodes prepared using sample RFC-300 (untreated and activated for 6 h) were evaluated with a self-made batch mode desalination apparatus. The change in concentration of solution was recorded during adsorption and desorption cycles and resultant profiles exhibited in Fig. 7. During desorption cycle at 0 V, concentration builds up to original value indicates nearly complete regeneration. In adsorption cycle, the concentration decreased from 500 to 292 ppm for untreated CA and to 88 ppm for CO₂ activated CA. The calculated salt-adsorption capacity according to equation 1 is 4.1 and 8.4 mg/g, respectively. The salt-adsorption capacity obtained for the CO₂ activated sample is significantly higher than the values reported earlier [24]. The synergetic effect of higher specific surface area and appropriate PSD is believed to be responsible for high electrosorption capacity. Fig. 7(b)

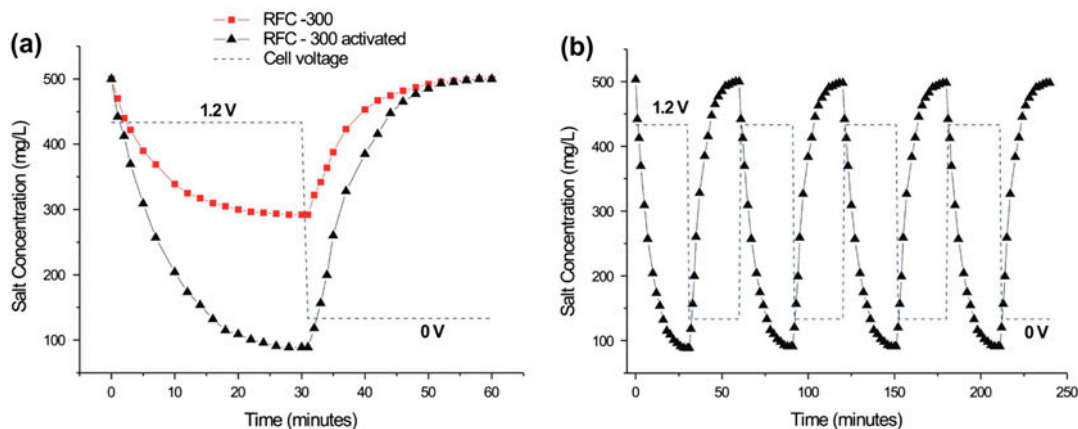


Fig. 7. (a) Concentration change in adsorption and desorption cycle (b) Sample performance in multiple cycles.

shows performance stability of sample in multiple cycles.

4. Conclusions

CAs were synthesized through polycondensation of resorcinol with formaldehyde by ambient drying method. By varying the catalyzing conditions of the precursor solution, the surface morphology and pore distribution could be tailored and CA with mesopores with average pore diameter of 9 nm was obtained. The surface textural properties of the CA generated were remarkably increased by CO₂ activation at 950 °C. The electrosorption behavior of prepared material was investigated by CV which displayed stable and superior specific capacitance after activation. The batch mode CDI experiments reveal that the material after activation has salt-adsorption capacity of 8.4 mg/g of active material. The enhanced electrosorption behavior of the activated CA electrode can be ascribed to well-controlled PSD with improved surface area.

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